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Fig. 1. Microstructure of an alloy with 90 wt.% (71.4 at.%) rhenium, 500 $\times$ . a –as-cast ( $\alpha + \sigma$ ); b –annealing at 1500 $^{\circ}$ , 5 h.  $\alpha + (\alpha + \beta)$

Figure 1: Fig. 1. Microstructure of an alloy with 90 wt.% (71.4 at.%) rhenium, 500 $\times$ . a –as-cast ( $\alpha + \sigma$ ); b –annealing at 1500 $^{\circ}$ , 5 h.  $\alpha + (\alpha + \beta)$

**Abstract**

**Full Text**

**Chemistry**

**M. A. TYLKINA, K. B. POVAROVA, and E. M. SAVITSKII**

## THE SIGMA PHASE IN THE RHENIUM-VANADIUM SYSTEM

*(Presented by Academician I. P. Bardin, 2 XII 1959)*

The sigma phase, as is known, is a metallic compound with a structure isomorphic to  $\beta$ -U and has a tetragonal lattice with 30 atoms in the unit cell. The sigma phase occurs in alloys of transition metals of the VA-VIA subgroups of D. I. Mendeleev's periodic system with transition metals of groups VIIA and VIII. Sigma phases are characterized by high hardness—of the order of 1200–1500 kg/mm<sup>2</sup> by Vickers.

**Fig. 1.** Microstructure of an alloy with 90 wt.% (71.4 at.%) rhenium, 500 $\times$ . a –as-cast ( $\alpha + \sigma$ ); b –annealing at 1500 $^{\circ}$ , 5 h.  $\alpha + (\alpha + \beta)$

According to literature data and our experimental data, rhenium (a metal of the VIIA subgroup) forms  $\sigma$ -phases with transition metals of the IVA subgroup—zirconium; of the VA subgroup—niobium, tantalum; of the VIA subgroup—chromium, molybdenum, tungsten; of the VIIA subgroup—manganese; and of group VIII—iron, with the size factor varying from +16.6% (Zr) to –8% (Fe). For  $\sigma$ -phases in systems with rhenium, an electron concentration of about 6.6–7.6 electrons/atom is characteristic (<sup>1–9</sup>). On the basis of the above data, especially with respect to Nb and Ta, it could be assumed that rhenium should also form a  $\sigma$ -phase with the element of the VA subgroup—vanadium; however, according to the data of work (<sup>1</sup>), no intermediate phases were found in the rhenium-vanadium system.

As a result of our work on constructing the phase diagram of the vanadium-rhenium system (<sup>10</sup>), it was established that the system contains a  $\sigma$ -phase with a narrow homogeneity range at a rhenium content of ~75 at.% (VRe<sub>3</sub>). This phase forms by a peritectic reaction between the melt and the rhenium-based

solid solution at 2490°. The microhardness of the  $\sigma$ -phase, measured on a PMT-3 instrument under a load of 100 g, is  $\sim 2000$  kg/mm<sup>2</sup>.

To establish the temperature interval of existence of the  $\sigma$ -phase, a series of high-temperature anneals of cast alloys was carried out (1750°, 7 h; 1500°, 5 h; 1000°, 450 h). The X-ray study was conducted in an RKD-type camera using CrK $\alpha$  radiation. According to X-ray structural and microstructural studies, eutectoid decomposition of the  $\sigma$ -phase at 1500° was established, with formation of a mixture of two solid solutions: vanadium-based ( $\alpha$ ) and rhenium-based ( $\beta$ ) (Fig. 1a, b).

On the X-ray pattern for the cast alloy, a system of lines characteristic of  $\sigma$ -phases is observed (Table 1); calculation of the lattice parameters gave the values  $a = 9.36$  Å,  $c = 4.86$  Å,  $c/a = 0.52$ .

**Table 1**

<i>hkl</i>	Intensity	Zr–Re		V–Re		Nb–Re		Ta–Re	
		$d \times \sin \theta$	$d$ Å	$d \times \sin \theta$	$d$ Å	$d \times \sin \theta$	$d$ Å	$d \times \sin \theta$	$d$ Å
112	s.	0,4483	2,340	0,5030	2,309	0,541	2,294	0,4884	2,305
410									
330	med.	0,4658	2,257	0,5180	2,243	0,551	2,247	0,5046	2,260
202	v.	0,4775	2,204	0,5299	2,193	0,563	2,202	0,5108	2,195
	weak								
212	weak	0,4909	2,163	0,5439	2,137	–	–	0,5294	2,132
420									
411	s.	0,5109	2,062	0,5548	2,093	0,590	2,098	0,5377	2,101
331	med.	0,5215	2,021	0,5678	2,048	–	–	0,5655	2,042
$a$ , Å		10,12		9,36		8,75		9,67	
$c$ , Å		5,42		4,86		4,06		4,97	
$c/a$		0,535		0,52		0,52		0,52	

<i>hkl</i>	Intensity	Cr–Re		Mo–Re		W–Re		Mn–Re		Fe–Re	
		$d \times \sin \theta$	$d$ Å	$d \times \sin \theta$	$d$ Å	$d \times \sin \theta$	$d$ Å	$d \times \sin \theta$	$d$ Å	$d \times \sin \theta$	$d$ Å
112	s.	0,5120	2,316	0,4957	2,308	0,497	2,308	0,5169	2,309	0,5234	2,311
410											
330	med.	0,5275	2,247	0,5089	2,247	0,512	2,250	0,5312	2,248	0,5390	2,245
202	v.	0,5385	2,202	0,5194	2,202	0,524	2,189	0,5416	2,205	0,5505	2,197
	weak										

<i>hkl</i>	Intensity	Cr–Re			W–Re			Mn–Re		Fe–Re	
		$d \times \sin \theta$	$d$	$\sin \theta$	$d \times \sin \theta$	$d$	$\sin \theta$	$d \times \sin \theta$	$d$	$\sin \theta$	$d \times \sin \theta$
212	weak	0,5530	2,143	0,5338	2,142	0,535	2,143	0,5569	2,143	0,5644	2,144
420											
411	s.	0,5675	2,090	0,5454	2,097	0,547	2,097	0,5707	2,092	0,5757	2,101
331	med.	0,5800	2,044	0,5594	2,044	0,560	2,048	0,5858	2,039	0,5919	2,045
<i>a</i> ,		9,23		9,54		9,54		9,14		9,02	
$\text{\AA}$											
<i>c</i> ,		4,80		4,96		4,97		4,75		4,69	
$\text{\AA}$											
<i>c/a</i>		0,52		0,52		0,52		0,52		0,52	

Table 1 presents comparative data from the calculation of X-ray patterns of  $\sigma$ -phases with CrK $\alpha$  radiation in the systems of rhenium with zirconium (<sup>4</sup>), vanadium, niobium (<sup>5</sup>), tantalum, chromium (<sup>6</sup>), molybdenum (<sup>7,8</sup>), tungsten (<sup>9</sup>), manganese, and iron (<sup>8</sup>).

On comparison, attention is drawn to a certain difference of the phase in the Zr–Re system, which we assigned to a type related to  $\sigma$ -phases, from the other  $\sigma$ -phases. This difference is probably explained by the fact that formation of  $\sigma$ -phases is not characteristic of metals of subgroup IVA, and the appearance of a  $\sigma$ -phase in the rhenium–zirconium system may be regarded as an exception.

In addition, an interesting fact is the formation of  $\sigma$ -phases in the system of rhenium with manganese and iron, elements of groups VIIA and VII (<sup>8</sup>), which indi-

icates the anomalous behavior of rhenium in comparison with metals of other groups.

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*Note: Figure translations are in progress. See original paper for figures.*

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