
AI translation · View original & related papers at
chinaxiv.org/items/chinaxiv-201611.00309

First-Principles Study of Alloying Effects on the Oxidation Resistance of γ -TiAl (Postprint)

Authors: Ping Faping, Hu Qingmiao, Yang Rui

Date: 2016-11-04T00:00:00+00:00

Abstract

Using the first-principles plane-wave pseudopotential method, we calculated the oxidation energies of Al_2O_3 and TiO_2 containing different transition-group alloying atoms and analyzed the influence of alloying on the relative stability of Al_2O_3 and TiO_2 . The computational results indicate that almost all alloying elements increase the oxidation energies of Al_2O_3 and TiO_2 , thereby decreasing their stability. The difference in oxidation energies between Al_2O_3 and TiO_2 demonstrates that Nb, Mo, W, Re, etc., significantly reduce the stability of Al_2O_3 relative to TiO_2 ; therefore, they can suppress the internal oxidation of the Al constituent in γ -TiAl and improve the high-temperature oxidation resistance of γ -TiAl.

Full Text

Investigation on Effects of Alloying on Oxidation Resistance of γ -TiAl by Using First Principles

Ping Faping, Hu Qingmiao, Yang Rui

Shenyang National Laboratory for Materials Science, Institute of Metal Research, Chinese Academy of Sciences, Shenyang 110016

Correspondent: Hu Qingmiao, professor, Tel: (024)23971813, E-mail: qmhu@imr.ac.cn

Supported by National Basic Research Program of China (No. 2011CB606404)

Manuscript received 2012-11-21, in revised form 2013-01-09

Abstract

The oxidation energies of Al_2O_3 and TiO_2 containing different transition metal alloying elements were calculated using a first-principles plane-wave pseudopo-

tential method, and the effect of alloying on the relative stabilities of Al₂O₃ and TiO₂ was also analyzed. The results showed that almost all the alloying elements increased the oxidation energies of Al₂O₃ and TiO₂, i.e., destabilized both Al₂O₃ and TiO₂. Comparing the oxidation energies of Al₂O₃ and TiO₂, it was found that W, Mo, Re, Nb, etc., decreased significantly the stabilities of Al₂O₃ relative to that of TiO₂, indicating that these alloying elements may hamper efficiently the inner oxidation of Al in the γ -TiAl matrix so as to increase the high-temperature oxidation resistance of γ -TiAl.

KEY WORDS first principle, γ -TiAl, alloying, oxidation energy, relative stability, internal oxidation

γ -TiAl alloys have attracted extensive attention as lightweight high-temperature structural materials due to their excellent specific strength and creep resistance at elevated temperatures (*cid* : 24). However, poor oxidation resistance at high temperatures limits their practical applications. The oxidation behavior of TiAl alloys is complex, involving the competitive formation of Al₂O₃ and TiO₂ scales. While a continuous Al₂O₃ layer provides effective oxidation protection, the simultaneous growth of TiO₂ compromises this protective effect.

Previous experimental studies have shown that alloying additions can significantly influence the oxidation resistance of γ -TiAl. Elements such as Nb, W, Mo, and Re have been reported to improve oxidation resistance, while others like Cr may have detrimental effects. The mechanisms behind these alloying effects remain incompletely understood, particularly regarding how different elements affect the relative stability of Al₂O₃ versus TiO₂ during the oxidation process.

In this work, we employ first-principles calculations to systematically investigate the effects of various transition metal alloying elements on the oxidation energies of Al₂O₃ and TiO₂. By comparing the relative stabilities of these oxides in the presence of different alloying elements, we aim to provide theoretical guidance for designing γ -TiAl alloys with improved oxidation resistance.

[Figure 1: see original paper] Overview of the effects of the alloy elements on the oxidation resistance of γ -TiAl [7–11,13,15]

The oxidation process in γ -TiAl involves the reaction of Al and Ti with oxygen to form Al₂O₃ and TiO₂. The relative stability of these oxides determines the protective nature of the oxide scale. According to thermodynamic principles, the formation of a continuous external Al₂O₃ layer (external oxidation) is preferred over internal oxidation of Al, which leads to mixed oxide formation and poor protection. The competition between Al₂O₃ and TiO₂ formation can be described by the reaction:

\$\$

The relative stability is governed by the Gibbs free energy change (ΔG) of this

reaction. A negative ΔG favors Al₂O₃ formation, while a positive value indicates preferential TiO₂ formation.

Computational Methodology

First-principles calculations were performed using the plane-wave pseudopotential method based on density functional theory (DFT). The generalized gradient approximation (GGA) with the Perdew-Burke-Ernzerhof (PBE) functional was employed for exchange-correlation effects. Ultrasoft pseudopotentials were used to describe electron-ion interactions, with a plane-wave cutoff energy of 400 eV. The k-point mesh was set to ensure energy convergence within 10⁻⁵ eV per atom, and forces on all relaxed atoms were converged to less than 0.1 eV/nm.

Supercell models were constructed for both Al₂O₃ and TiO₂ with substitutional alloying elements. For Al₂O₃, a 2×2×1 supercell containing 24 Al and 36 O atoms was used, while for TiO₂ a 3×3×1 supercell with 36 Ti and 72 O atoms was employed. Transition metal elements (Nb, Mo, W, Re, etc.) were substituted at cation sites to simulate alloying effects.

Results and Discussion

Structural Properties The calculated equilibrium lattice parameters for TiO₂ and Al₂O₃ are summarized in Tables 1 and 2, respectively. The results show good agreement with experimental data and previous theoretical calculations, validating our computational approach.

Calculated equilibrium lattice parameters of TiO₂ in comparison with experimental and other theoretical data

Calculated equilibrium lattice parameters of Al₂O₃ in comparison with experimental and other theoretical data

Oxidation Energies The oxidation energy (ΔH) for each oxide system was calculated as the energy difference between the oxidized state and the reference states of the constituent elements. The effect of alloying elements on the oxidation energies of Al₂O₃ and TiO₂ is shown in Figure 2.

[Figure 2: see original paper] Effect of alloying on the oxidation energies of Al₂O₃ (a) and TiO₂ (b)

The results demonstrate that almost all alloying elements increase the oxidation energies of both Al₂O₃ and TiO₂, indicating a destabilizing effect on both oxides. However, the degree of destabilization differs between the two oxides, which critically affects their relative stability.

Relative Stability The relative stability between Al₂O₃ and TiO₂ in the presence of alloying elements was evaluated by comparing their oxidation energies.

Figure 3 illustrates the change in relative stability ($\Delta\Delta H$) induced by different alloying elements.

[Figure 3: see original paper] Effect of alloying on the relative stability of Al₂O₃ and TiO₂

Elements such as W, Mo, Re, and Nb significantly decrease the relative stability of Al₂O₃ compared to TiO₂. This destabilization of Al₂O₃ relative to TiO₂ suggests that these elements may suppress the internal oxidation of Al in the Al-TiAl matrix. The mechanism involves the preferential interaction of these alloying elements with the oxide structure, altering the electronic states and bonding characteristics.

Electronic Structure Analysis To understand the underlying electronic mechanisms, we analyzed the density of states (DOS) for pure and doped oxide systems. Figure 4 shows the total and projected DOS for Al₂O₃ and TiO₂.

[Figure 4: see original paper] Total and atomic projected densities of states for Al₂O₃ (a) and TiO₂ (b)

The valence band of Al₂O₃ is dominated by O 2p states, while the conduction band consists primarily of Al 3s and 3p states. For TiO₂, the valence band comprises O 2p states hybridized with Ti 3d states, and the conduction band is mainly Ti 3d character. The band gap of Al₂O₃ is approximately 6.0 eV (GGA value, underestimated compared to the experimental ~8.7 eV), while TiO₂ shows a gap of about 1.7 eV.

Charge Transfer Analysis Bader charge analysis was performed to investigate the charge redistribution induced by alloying elements. Table 3 presents the charge transfer effects of W and Cu doping in Al₂O₃ and TiO₂.

Effects of W and Cu on the charge transfer in Al₂O₃ and TiO₂

In pure Al₂O₃, Al loses electrons (+2.481 e) and O gains electrons (-1.654 e). In pure TiO₂, Ti transfers +2.280 e to O (-1.136 e). When W is doped into Al₂O₃, the charge on Al changes slightly to +2.478/+2.480 e, while O becomes less negative (-1.545/-1.549 e). The W dopant itself carries a positive charge of +1.825 e, indicating electron transfer to neighboring O atoms.

In TiO₂, W doping reduces the positive charge on Ti from +2.280 to +2.195 e, while the O charge becomes more negative (-1.395 e). The W dopant carries a significant positive charge of +2.785 e, suggesting strong interaction with oxygen. This enhanced charge transfer correlates with the observed destabilization of TiO₂ relative to Al₂O₃.

The difference in charge transfer behavior between W and Cu is notable. Cu doping results in less pronounced charge redistribution, with Cu carrying only +1.301 e in TiO₂, which correlates with its weaker effect on oxidation resistance compared to W.

Mechanism of Alloying Effects The enhanced oxidation resistance by elements like W, Mo, Re, and Nb can be explained by their influence on the competitive oxidation process. These elements destabilize Al₂O₃ more significantly than TiO₂, which may seem counterintuitive since Al₂O₃ is the desired protective scale. However, this relative destabilization actually suppresses the internal oxidation of Al, promoting the formation of a continuous external Al₂O₃ layer rather than dispersed internal oxide precipitates.

The schematic in Figure 6 illustrates the difference between internal and external oxidation morphologies. Internal oxidation occurs when oxygen dissolves in the alloy and reacts with Al internally, forming discrete Al₂O₃ particles. External oxidation forms a continuous surface layer that provides better protection.

[Figure 6: see original paper] Schematic morphology of Al₂O₃ in different growth models: (a) internal oxidation, (b) external oxidation

The key to improving oxidation resistance is promoting external oxidation at the expense of internal oxidation. Elements that destabilize Al₂O₃ relative to TiO₂ reduce the driving force for internal Al₂O₃ formation, thereby favoring external scale growth when the overall Al activity is sufficient.

Conclusion

First-principles calculations reveal that transition metal alloying elements W, Mo, Re, and Nb significantly decrease the relative stability of Al₂O₃ compared to TiO₂. This selective destabilization suppresses internal oxidation of Al in α -TiAl alloys, thereby promoting the formation of protective external Al₂O₃ scales and improving high-temperature oxidation resistance. The electronic mechanism involves modified charge transfer and bonding characteristics in the oxide structures. These theoretical findings provide guidance for designing oxidation-resistant α -TiAl alloys through strategic alloying additions.

References

- [1] Clemens H, Kestler H. *Adv Eng Mater*, 2000; 2: 551
- [2] Kim Y W. *JOM*, 1989; 41: 24
- [3] Subramanian P R, Mendiratta M G, Dimiduk D M, Stucke M A. *Mater Sci Eng*, 1997; A239-240: 1
- [4] Becker S, Rahmel A, Schorr M, Schutze M. *Oxid Met*, 1997; 48: 425
- [5] Brady M, Brindley W, Smialek J, Locci I. *JOM*, 1996; 48: 46
- [7] Shida Y, Anada H. *Mater Sci Eng*, 1993; A172: 131
- [9] Rahmel A, Schutze M, Quadakkers W J. *Mater Corros*, 1995; 46: 271
- [11] Kekare S A, Aswath P B. *Oxid Met*, 1995; 43: 165
- [13] Brady M P, Smialek J L, Humphrey D L. *Oxid Met*, 1997; 48: 71
- [14] Kekare S A, Aswath P B. *Oxid Met*, 1995; 43: 165
- [15] Dettenwanger F, Schumann E, Rakowski J M, Meier G H, Ruhle M. *Oxid Met*, 1998; 50: 269
- [21] Rakowski J M, Meier G H, Ruhle M. *Oxid Met*, 1998; 50: 269

- [22] Schumann E, Dettenwanger F, Rakowski J M, Meier G H, Ruhle M. Mater Corros, 1997; 48: 38
- [28] Wang L, Maxisch T, Ceder G. Phys Rev, 2006; 73B: 195107
- [29] Mo S D, Ching W Y. Phys Rev, 1995; 51B: 13023
- [30] Burdett J K, Hughbanks T, Miller G J, Richardson J W, Smith J V. J Am Chem Soc, 1987; 109: 3639
- [31] Martinez J I, Hansen H A, Rossmeisl J, Norskov J K. Phys Rev, 2009; 79B: 045120
- [32] Boettger J C. Phys Rev, 1997; 55B: 750
- [33] Damour H, Schiferl D, Denner W, Schulz H, Holzapfel W B. J Appl Phys, 1978; 49: 4411
- [34] Li M S. High Temperature Corrosion of Metals. Beijing: Metallurgical Industry Press, 2001: 11
- [35] Wang L, Maxisch T, Ceder G. Phys Rev, 2006; 73B: 195107
- [36] Martinez J I, Hansen H A, Rossmeisl J, Norskov J K. Phys Rev, 2009; 79B: 045120
- [37] Sousa C, Illas F. Phys Rev, 1994; 50B: 13974
- [38] Marquez A M, Plata J J, Ortega Y, Sanz J F. J Phys Chem, 1992; 38: 425

Note: Figure translations are in progress. See original paper for figures.

Source: ChinaXiv –Machine translation. Verify with original.